

# Understanding and modeling of the ply-ply friction response for UD C/PAEK tapes in melt

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#### UNDERSTANDING AND MODELING PLY-PLY FRICTION

#### Contents:

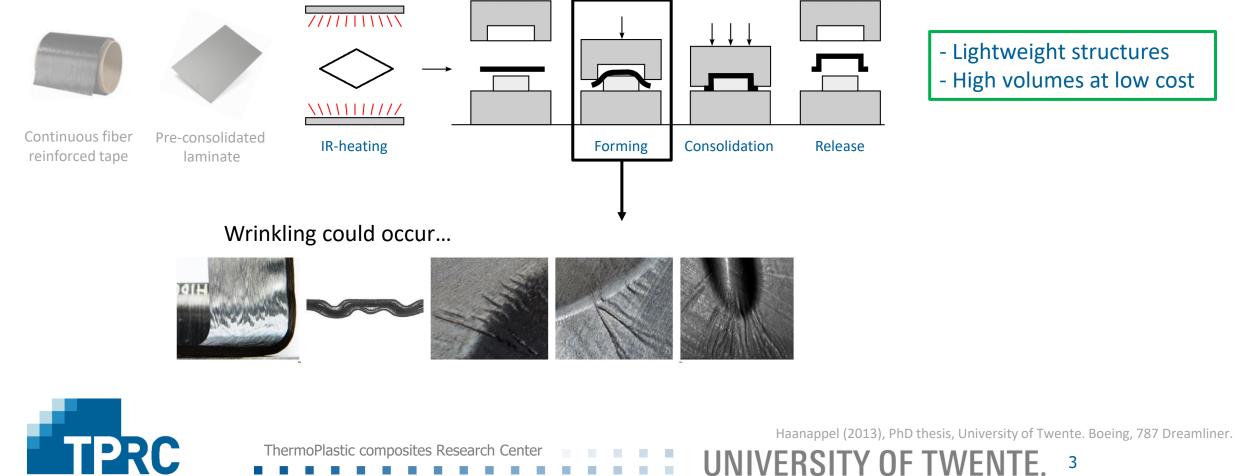
- Introduction hot press forming and ply-ply friction
- Experimental work
- Modeling:
  - Peak and long-time shear stress
  - Transient friction
    curve
- Conclusion

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#### **INTRODUCTION**

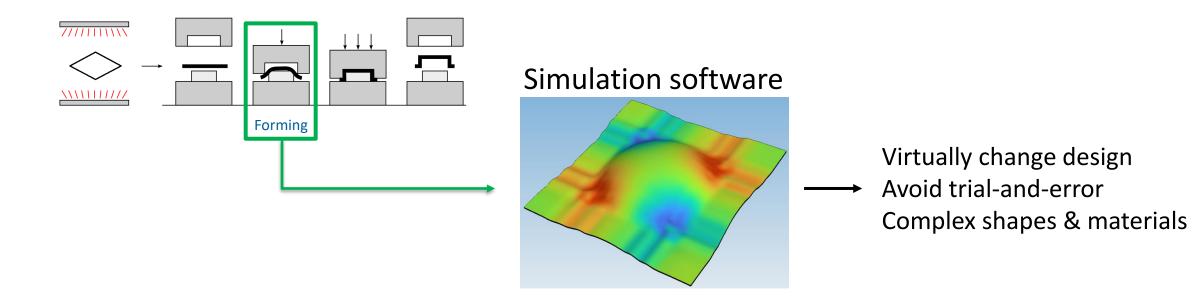
Manufacturing composite parts

Hot press forming of thermoplastic composites: 





Simulation software



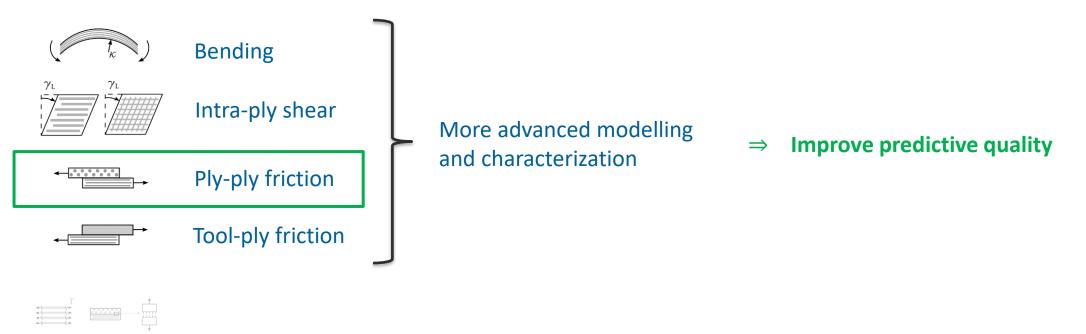
But predictions need to be accurate... especially problem for UD reinforcements



#### INTRODUCTION

Strategy

- Process simulation software:
  - Mathematical description of deformation mechanisms



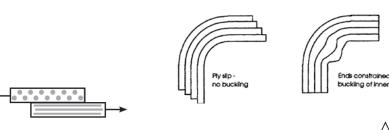


Sachs (2014). Friction and bending in thermoplastic composites forming processes

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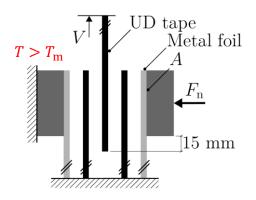
### INTRODUCTION

Ply-ply friction



- Situation: ply-ply friction
  - Relative movement between successive plies
  - Friction tester to characterize response

relipbuckling Ends constrained , no stpbuckling of inner piles more than two fibre directions fibre directions inter-ply slip

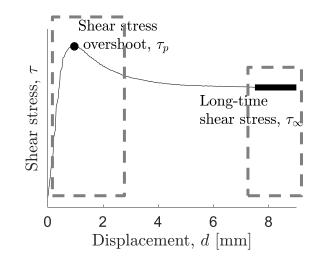


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- Complication:
  - Often only  $\tau_{\infty}$  considered...
  - Most slip lengths in simulations correspond to transient regime
- Questions:

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- What is the underlying mechanism for the friction response?
- Can we model the transient friction response?

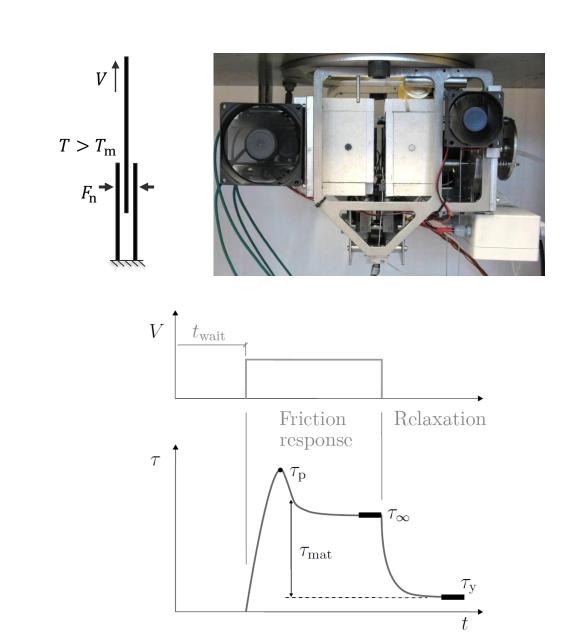


Sachs (2014). PhD thesis, University of Twente. Murtagh et al (1995). Comp. Manuf. 6, 169-175. Haanappel (2013). PhD thesis, University of Twente.

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Ply-ply friction tests

- Material:
  - Toray TC1200 UD C/PEEK (59% V<sub>f</sub>)
  - Processing T: 370-400 °C
  - 0°/0° lay-up in specimen
- Experimental conditions:
  - Constant T of 385 °C and p of 15 kPa
  - Range of sliding rates V: 1-200 mm/min
  - 5 minute heating time



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Ply-ply friction tests

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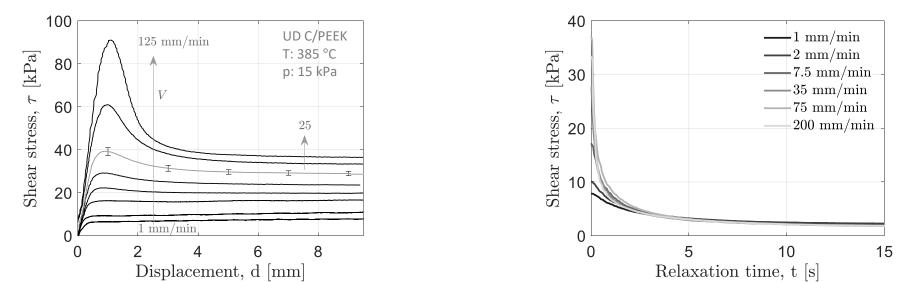
- Friction response:
  - Peak increases with increasing sliding rate V
  - $\tau_{\infty}$  bounded at limiting value



 Stress drops after sliding action stopped

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Residual after 'full' relaxation

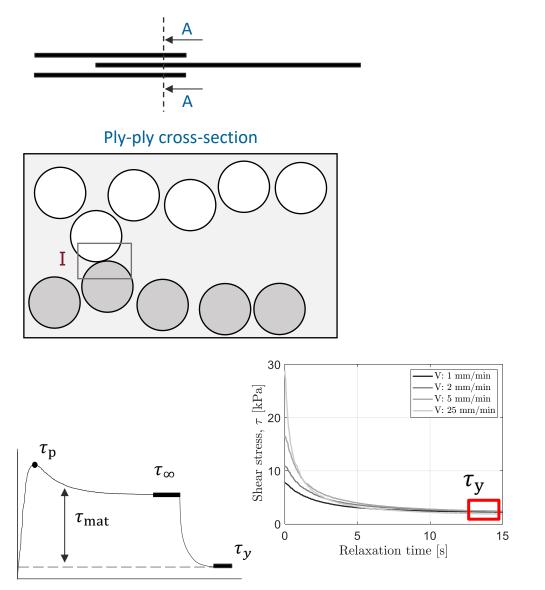


Pierik et al. (2021). Proceedings of 24<sup>th</sup> International Conference of Material Forming, Liege, Belgium, 3695.

Fiber-fiber interaction

- Ply-ply cross-section:
  - Matrix interlayer in melt
  - Fiber-fiber contact
- Fiber interactions:

- Could result in residual stress
- Distinguish VE matrix response  $\tau_{mat}$  and static yield stress  $\tau_{v}$  from total response



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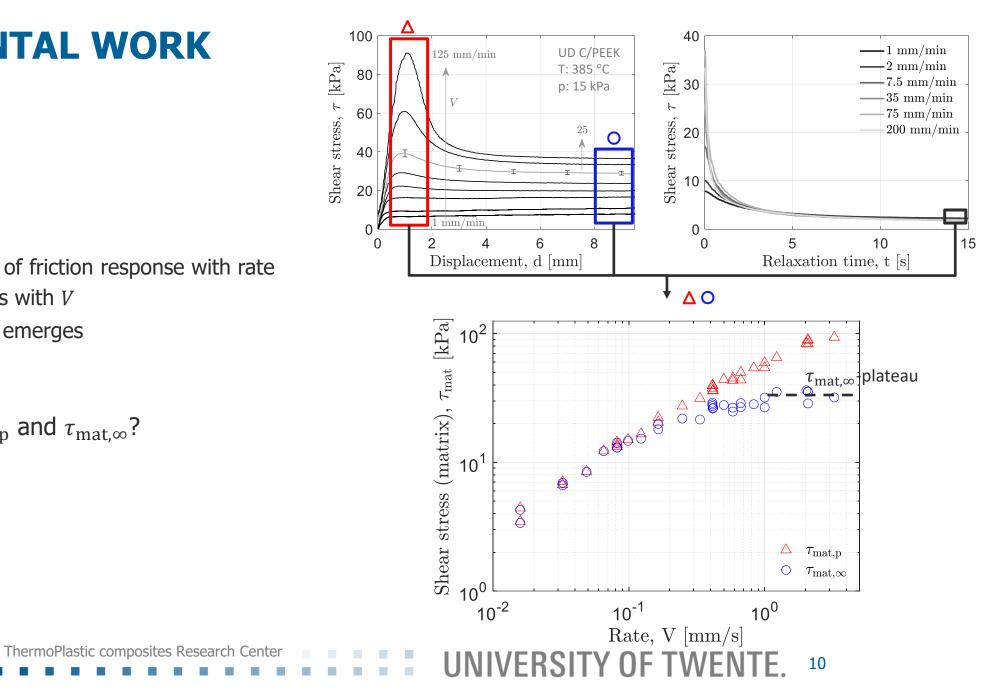
Murtagh et al. (1993), in: A. Miravete (Ed.), Proceedings of ICCM-9, Madrid, Spain, pp 311-318. Cogswell (1992). Thermoplastic Aromatic Polymer Composites, UK: Butterworth-Heinemann. Advani et al. (1997), Chapter 8 in: D. Bhattacharyya (Ed.), Composite Materials Series, vol. 11, Elsevier, pp 323-369.

Flow curve

Flow curve: 

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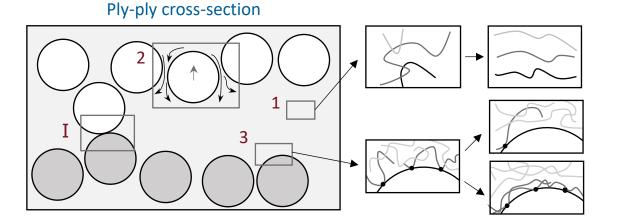
- Characteristics of friction response with rate
- $\tau_{mat,p}$  increases with V
- $\tau_{mat,\infty}$ -plateau emerges
- What about  $\tau_{mat,p}$  and  $\tau_{mat,\infty}$ ?



Hypotheses

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- Hypotheses for transient (matrix) response:
  - 1. Nonlinear viscoelasticity
  - 2. Increase matrix interlayer thickness
  - 3. Slip relaxation effect (wall slip)



• All revolve around some kind of shearing action in matrix interlayer at ply-ply interface...



Shear flow model

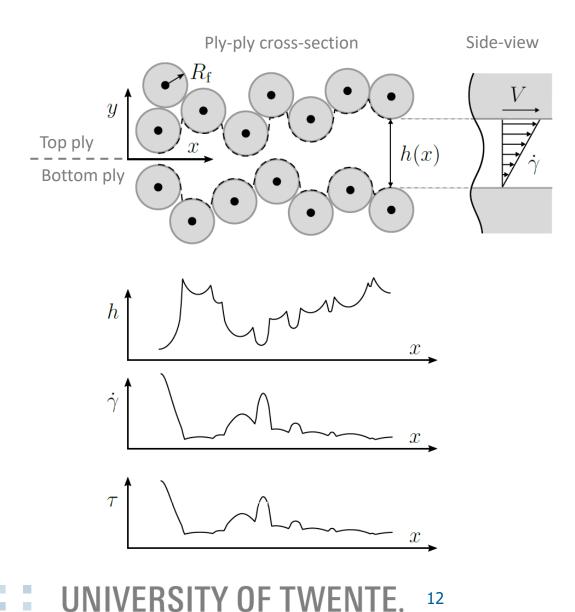
- Model friction at ply-ply interface:
  - Assuming 1D shear flow of matrix material
  - No interference of flow in width direction
  - Matrix interlayer thickness distribution h(x)
  - Viscous flow:

• 
$$\tau_{\text{visc}}(x) = \eta \left(\frac{V}{h(x)}\right) \frac{V}{h(x)} \Rightarrow \tau_{\text{avg}} = \frac{1}{w} \int_0^w \tau_{\text{visc}}(x) \, dx$$

- Model inputs:
  - Matrix viscosity from plate-plate rheometry
  - Matrix interlayer thickness distribution h(x) from micrographs or generated fiber distr.

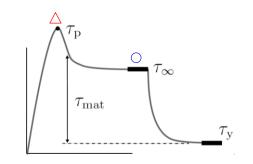
Pierik et al. (2022). Compos. Part A: App Sci and Manuf 163, 107185. Melro et al (2008). Compos. Sci. Tech., 68, 2092-2102.



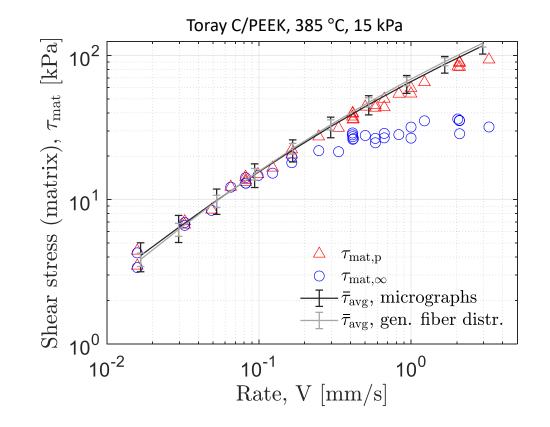


Comparison model and measurement

- Prediction of peak shear stress  $\tau_{mat,p}$ :
  - 16 (smoothed) h(x) from micrographs
  - 10 generated fiber distributions
- Good agreement model ( $\bar{\tau}_{avg}$ ) & measurement
- Recall hypotheses to model  $\tau_{mat,\infty}$ :
  - (Wall) slip relaxation effect most probable
  - Literature on polymer melts: rate jump (stress-contr.) or stress plateau (rate-contr.) at critical shear stress



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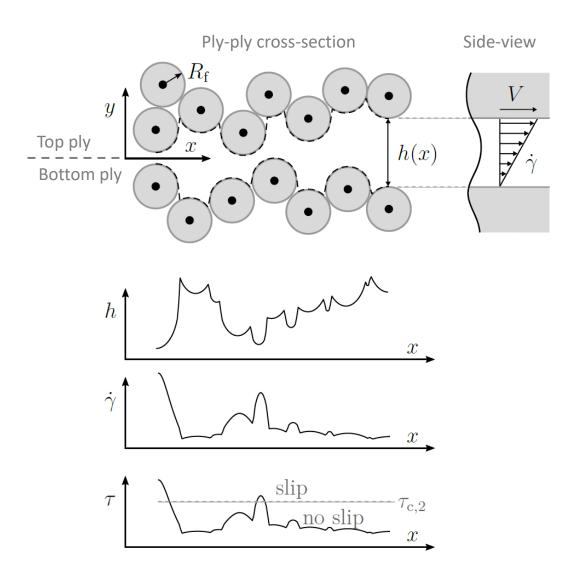


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Hatzikiriakos (2012) Prog Polym Sci 37, Boukany et al (2009) Macromolecules 42, Park et al (2008) J Rheol 52.

Shear flow model

- Recall the shear flow model:
  - Viscous flow of matrix material at ply-ply interface
- Include (strong) wall slip:
  - Use  $\tau_{c,2}$  as a limiting value
  - Bound viscous shear stress distribution:
    - $\tau_{\text{visc}}(x) = \min\left[\eta\left(\frac{V}{h(x)}\right)\frac{V}{h(x)}, \tau_{c,2}\right]$
  - Base  $\tau_{c,2}$  on experimental results:  $\tau_{mat,\infty}$ -plateau



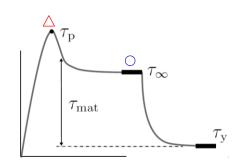
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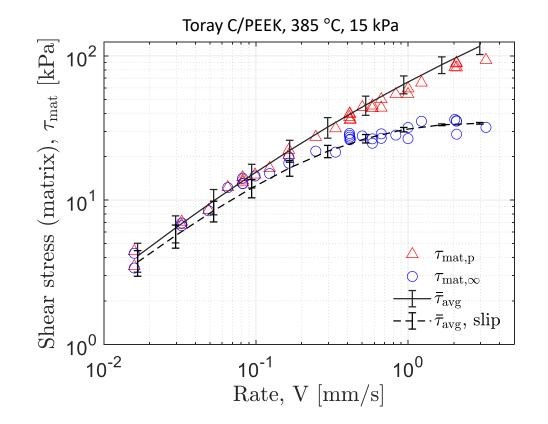
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Hatzikiriakos et al (1992) J Rheol 36, Boukany et al (2009) Macromolecules 42, Park et al (2008) J Rheol 52.

Shear flow model

- Good agreement modeled and measured long-time shear stress  $\tau_{mat,\infty}$
- Gradual transition from no-slip to full-slip with sliding rate due to h
- Combined experimental and modeling work substantiate concept of wall slip



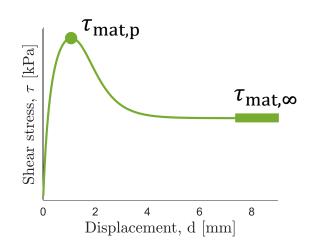


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Pierik et al. (2022). Compos. Part A: App Sci and Manuf 163, 107185.

The next step

- What we have:
  - Model for characteristics of friction response
- What we want:
  - Describe the full transient response, but no too computationally expensive...
- How?
  - Further explore the identified underlying mechanisms...



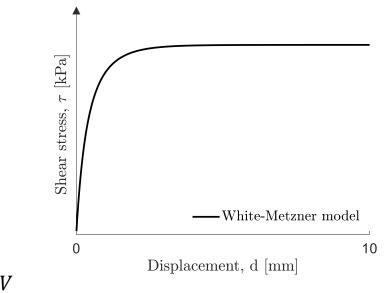


Conceptualization

- Proposed mechanisms:
  - 1. Shear flow of matrix material at ply-ply interface
  - 2. Wall slip relaxation effect at fiber-matrix interface
- 1. Shear flow of matrix material:
  - Viscoelastic model for stress growth with  $\eta(\dot{\gamma})$  and h(x)
  - White-Metzner model:

$$\tau_{12} + \lambda(\dot{\gamma}) \frac{\mathrm{d}\tau_{12}}{\mathrm{d}t} = \eta(\dot{\gamma})\dot{\gamma} \text{ with } \lambda(\dot{\gamma}) = \frac{\eta(\dot{\gamma})}{G_{\mathrm{WM}}} \text{ and } \dot{\gamma} = \frac{V}{h(x)}$$

• Adjust shear rate for wall slip

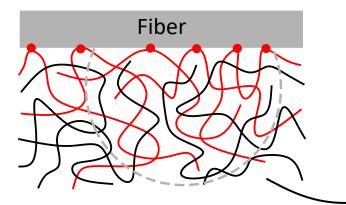


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White, Metzner (1983). J. of Appl. Polym. Sci., 7 (5), 1867-1889.

Conceptualization

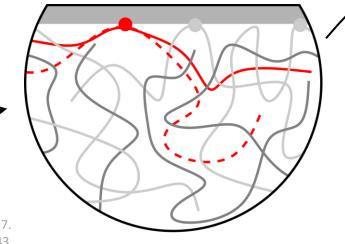
- 2. Wall slip relaxation effect:
  - Transition from no-slip towards fully-developed wall slip at high rates
  - Critical shear stress  $\tau_{c,2}$  for onset of wall slip
  - Proposed slip mechanism: disentanglement



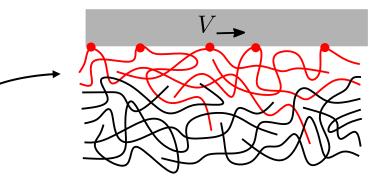
Rest state; surface chains (red) adsorbed to the wall (fiber) and entangled with bulk chains

Brochard, de Gennis (1992). Langmuir, 8 (12), 3033-3037. Hatzikiriakos (2012). Progr. in Polym. Sci., 37 (4), 624-643.

Surface chains elongate and retract from bulk due to applied shear flow



Surface chains settle to new eq. state with less bulk interaction

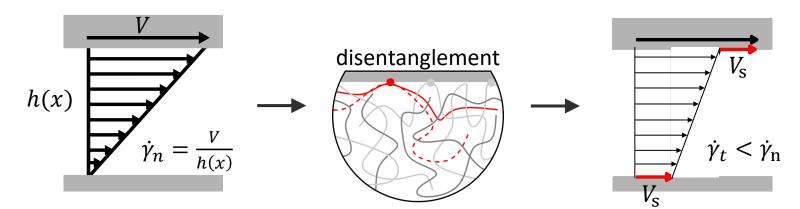


- Eq. state corresponding to applied shear flow
- Still dis- and re-entanglement
- Less interaction, thus less momentum transfer

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Conceptualization

- 2. Wall slip relaxation effect:
  - Slip velocity *V*<sub>s</sub> emerges with ongoing deformation



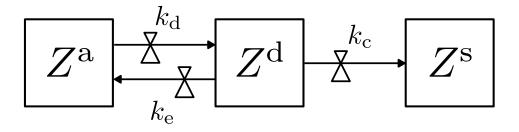
•  $V_{\rm s}$  to correct for the changing BC:  $\dot{\gamma}_{\rm t} = \frac{V - 2V_{\rm s}}{h(x)}$ 

 $\Rightarrow$  Thus, now we just need to describe  $V_{\rm s}(t)$ ...

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The approach

- Idea: describe process of disentanglement with state-rate model
- Consider a relative number of active surface chains  $Z^a$  in the matrix-fiber interface
  - 1. In rest state, all surface chains interact with bulk through entanglements
    - $\rightarrow Z^{a}$  equals unity
  - 2. Upon shear deformation, surface chains retract to disentangled state  $Z^{d}$  with rate  $k_{d}$ 
    - A disentangled surface chain could:
    - a. Relax and re-entangle with entanglement rate  $k_{\rm e}$
    - b. Settle into equilibrium state  $Z^s$  corresponding to applied shear deformation with conformation rate  $k_c$
  - 3. With  $t \to \infty$ , all surface chains settled and  $Z^s = 1$

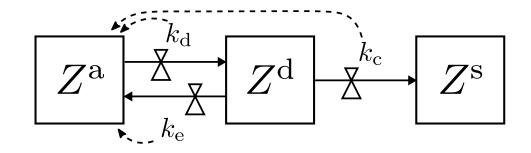


$$\frac{dZ^{a}}{dt} = -k_{d}Z^{a} + k_{e}Z^{d}$$
$$\frac{dZ^{d}}{dt} = k_{d}Z^{a} - k_{e}Z^{d} - k_{c}Z^{d}$$
$$\frac{dZ^{s}}{dt} = k_{c}Z^{d}$$

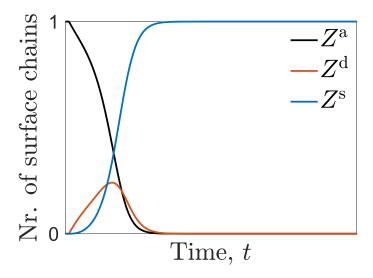
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State-rate model

- Rates  $k_i$  need to be determined...
  - Probably depend on states  $Z^i$  and V



- Use linear interpolation between  $k_{\min}$  and  $k_{\max} = k_{\min} + aV^m$  based on  $Z^a$ :
  - At start-up: low disentanglement  $k_{\rm d}$ , high entanglement  $k_{\rm e}$
  - Conformation k<sub>c</sub> same rate as k<sub>d</sub>





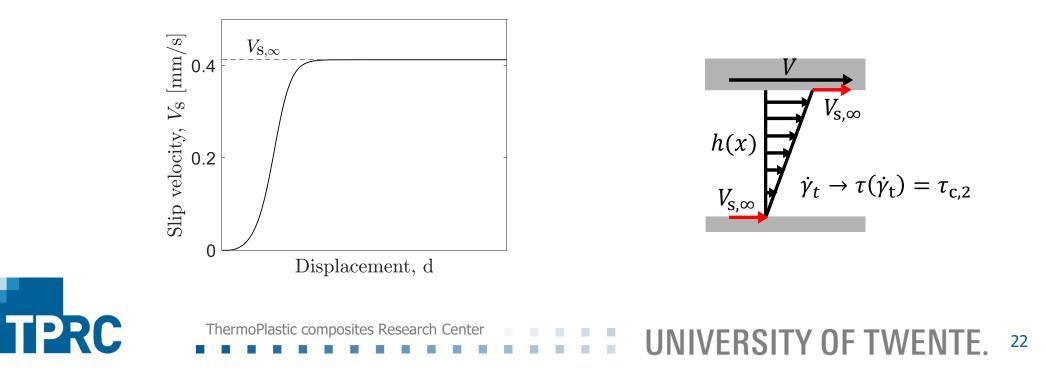
#### **STATE-RATE MODEL**

Implementation with  $V_{\rm s}$ 

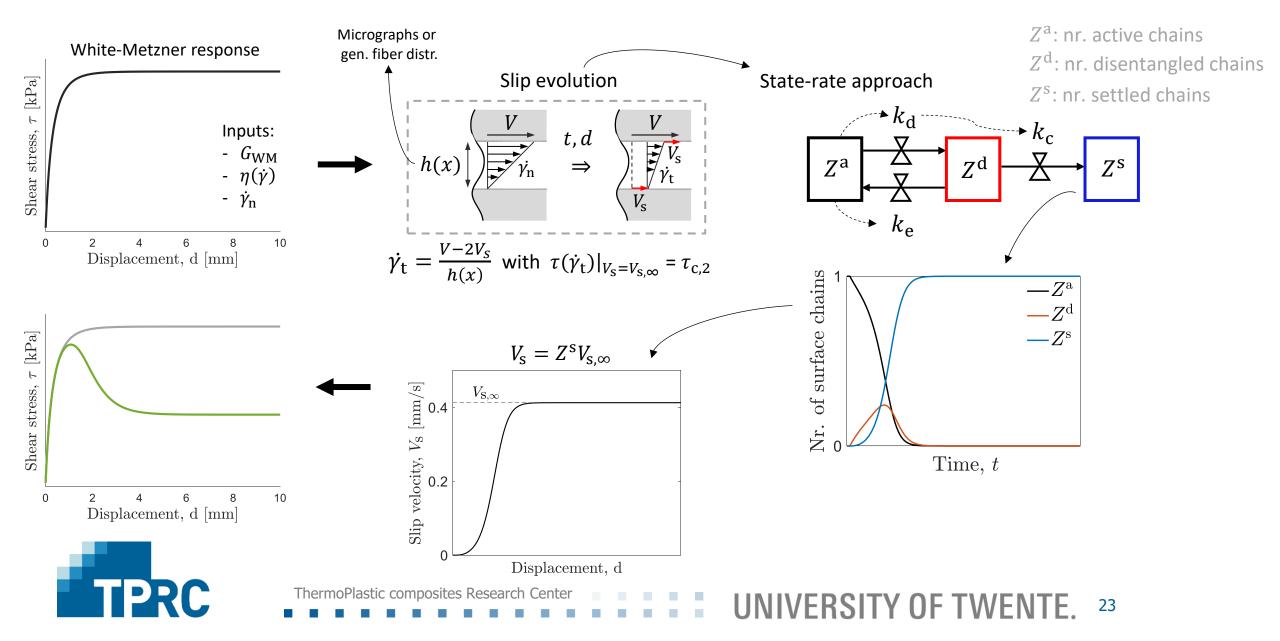
• Relate evolution of *Z*<sup>s</sup> to slip velocity:

$$V_{\rm s}(t)=V_{\rm s,\infty}Z^{\rm s}(t),$$

with  $V_{s,\infty}$  the slip velocity in steady-state (VE shear stress = interfacial shear stress =  $\tau_{c,2}$ )

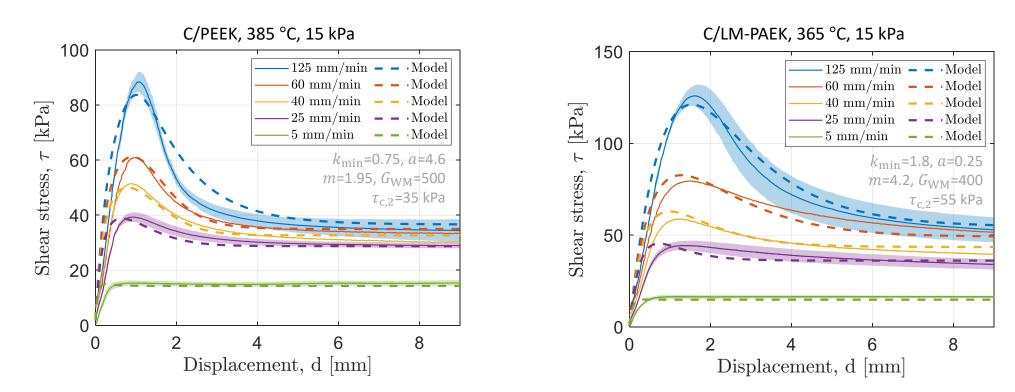


#### OVERVIEW FULL TRANSIENT MODELING



Model & measurement

• Comparison model and measurement:





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- Slip relaxation or gradual increase of wall slip explains the observed transient ply-ply friction response of UD TPC tape in melt, exhibiting a peak followed by steady-state friction
- The peak friction can be predicted by shear flow of matrix material at the ply-ply interface
- The steady-state friction can be predicted through including a critical shear stress for the onset of (strong) wall slip
- The full transient (start-up) friction response can be captured with VE framework combined with a state-rate model that describes the evolution of the slip velocity at the fiber-matrix interface
- Future work is to implement the proposed transient shear flow model in simulation software





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#### Thank you for your attention!

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