

Flow and Transient Phenomena in Large Scale Additive Manufacturing.

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FLOW PROCESSES IN COMPOSITE MATERIALS - 15





Agenda

- Large scale additive manufacturing
- Physics-based simulation workflow ADDITIVE3D for composites additive manufacturing
- Overview of physics considered
- Examples of simulation driven design with ADDITIVE3D
- Horizons and opportunities in composites additive manufacturing





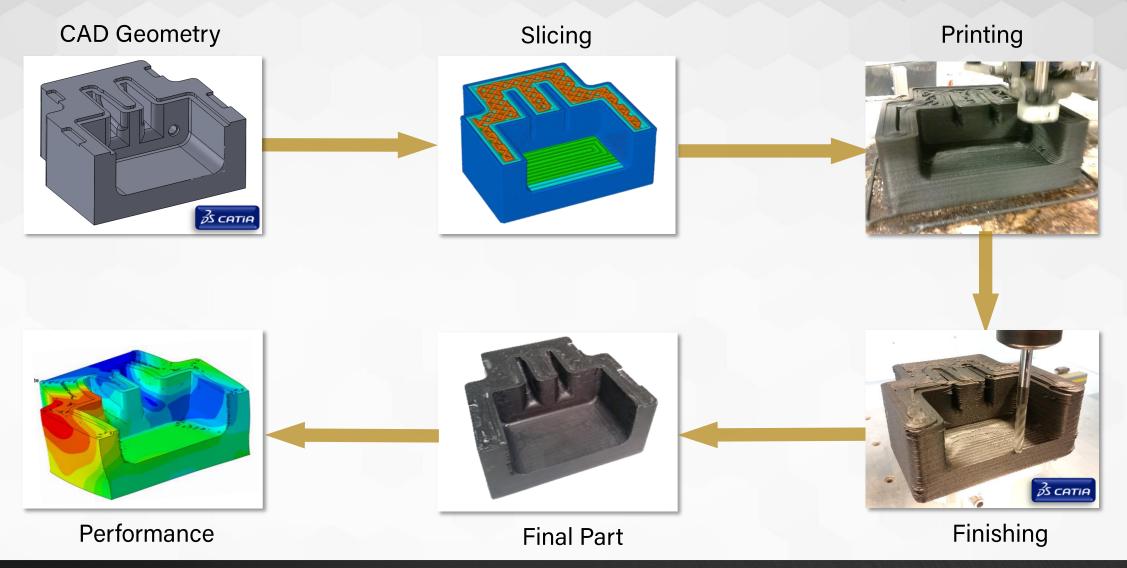
Large Scale Additive Manufacturing (LSAM)





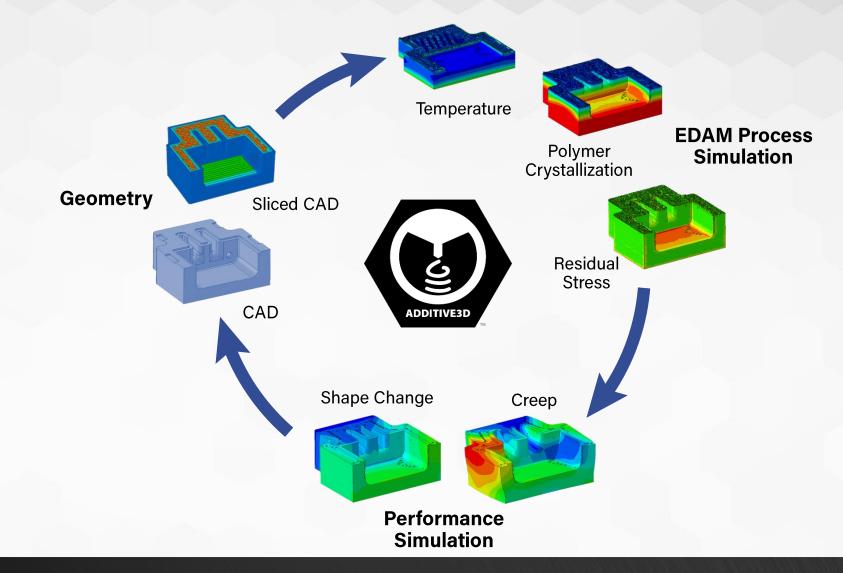


Workflow for Additive Manufacturing





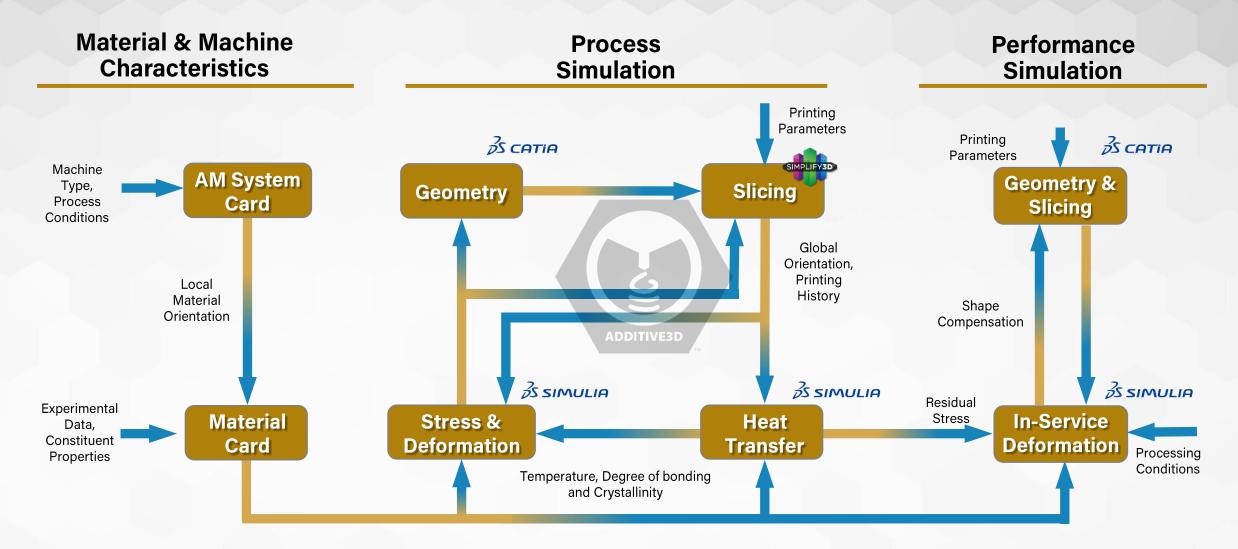
ADDITIVE3D: Simulation Workflow for AM





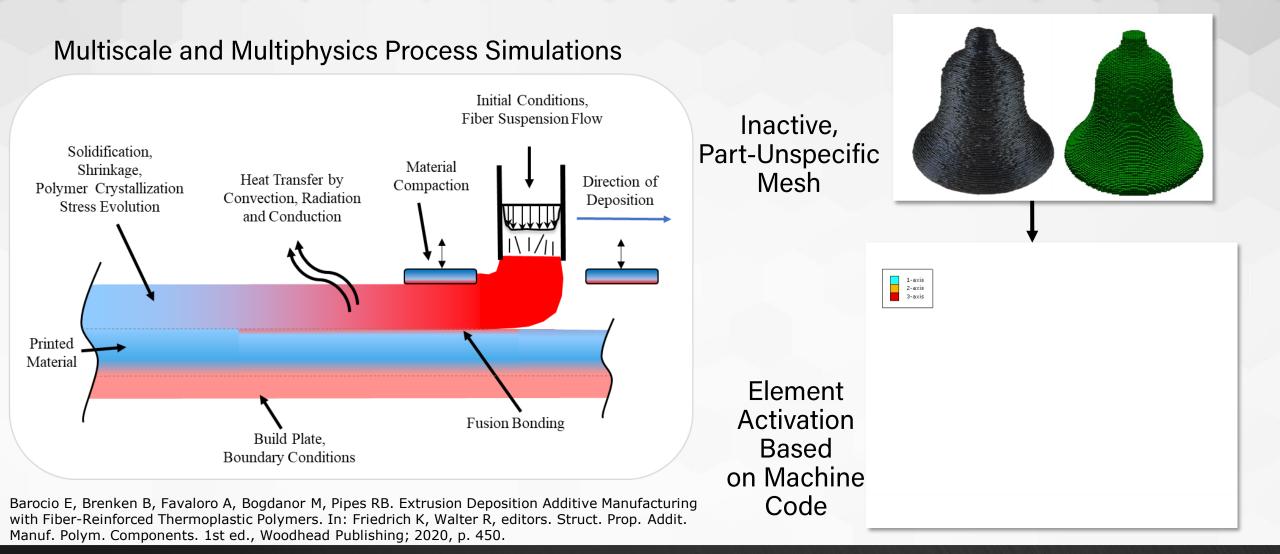


ADDITIVE3D[™] Process and Performance Simulation



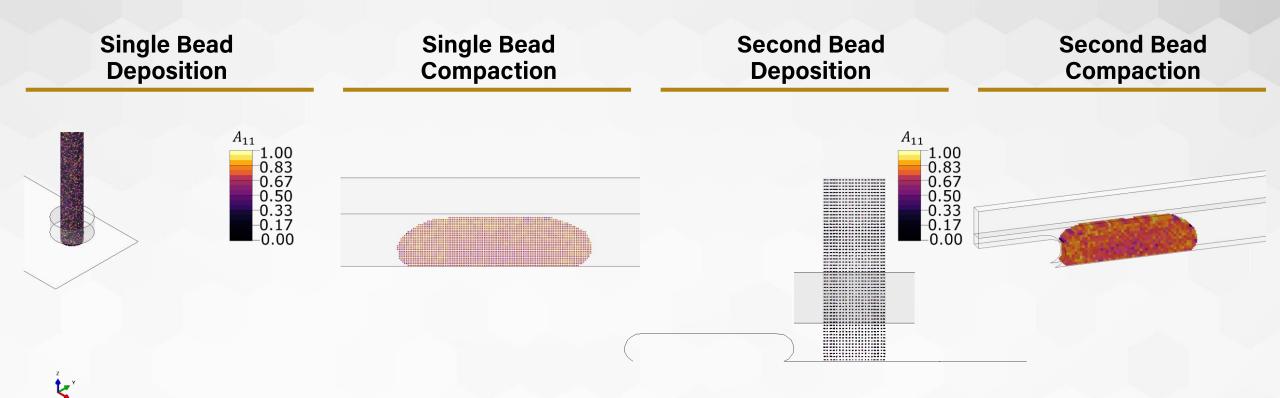


Physics Captured in AM Process Simulation





Anisotropic Flow of Fibers Polymer Suspension



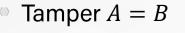
Pibulchinda P, Barocio E, Favaloro AJ, Pipes RB. Influence of printing conditions on the extrudate shape and fiber orientation in extrusion deposition additive manufacturing. Compos Part B Eng 2023;261:110793. https://doi.org/10.1016/j.compositesb.2023.110793.



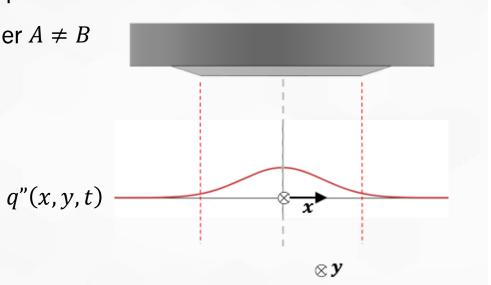
Heat Losses in Material Compaction

- Actively cooled compacter
- Modeled as a moving heat flux with double gaussian distribution

$$q''(x, y, t) = \frac{Q^C \sqrt{AB}}{\pi} e^{-A(x+v_x t)^2} e^{-B(y+v_y t)^2}$$



Roller $A \neq B$





* Convection and radiation disabled

cm=c

NT11

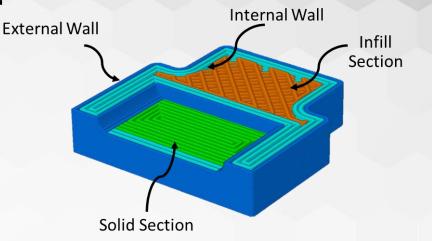
- 559. - 546. - 533. - 520. - 507. - 494. - 481. - 468.7 - 468.7 - 455.8 - 442.9 - 430.0

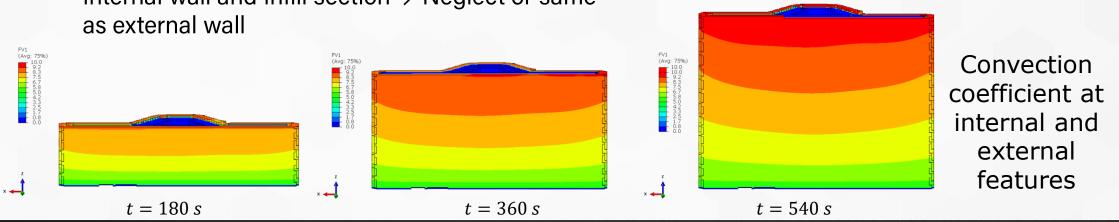


Local Convection Conditions

Feature dependent convection coefficients assumes:

- **Convection occurs instantaneously**
- Features are non-interactive
- Analogies made:
 - Solid section \rightarrow Horizontal surface
 - External wall \rightarrow Vertical surface
 - Internal wall and Infill section \rightarrow Neglect or same as external wall





Polymer Crystallization Kinetics

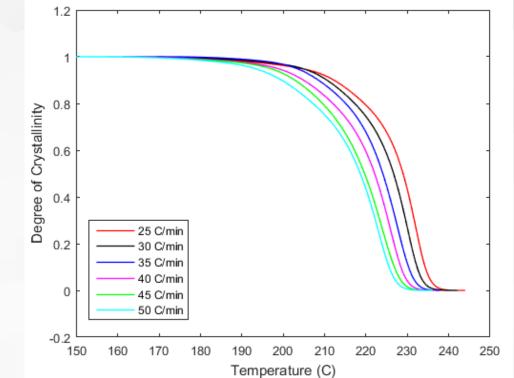
- Polymer crystallization is cooling rate dependent
- Dual crystallization mechanisms observed in CF-PPS →
 Crystallization model developed by Velisaris and Seferis

$$X_{vc}(T,t) = X_{vc\infty}(w_1 F_{vc_1} + w_2 F_{vc_2})$$

$$F_{\nu c_{i}} = 1 - \exp\left[-C_{1_{i}} \int_{0}^{t} T \cdot exp\left[\frac{-C_{2_{i}}}{(T - T_{g} + T_{c_{i}})} - \frac{C_{3_{i}}}{\left(T\left(T_{m_{i}} - T\right)^{2}\right)}\right] n_{i}\tau^{n_{i}-1}d\tau$$

$$i = 1,2$$

- X_{vc} crystallinity volume fraction, w_1 and w_2 are weight factors
- C_{1_i} pre-exponential factors capturing the temperature dependence
- C_{2_i} parameters associated with the temperature dependence of diffusion
- $\mathcal{C}_{\mathbf{3}_i}$ parameters associated with the free enthalpy of nucleation
- n_i Avrami coefficients for each crystallization mechanism



Barocio E, Brenken B, Favaloro A, Bogdanor M, Pipes RB. Extrusion Deposition Additive Manufacturing with Fiber-Reinforced Thermoplastic Polymers. In: Friedrich K, Walter R, editors. Struct. Prop. Addit. Manuf. Polym. Components. 1st ed., Woodhead Publishing; 2020, p. 450.

Polymer Melting

 Melting model assumes statistical distribution of lamellar thicknesses (Greco & Maffezzoli) -> Only temperature dependent

$$T_m = T_m^\circ - \frac{2\gamma_e T_m^\circ}{l\Delta H_f}$$

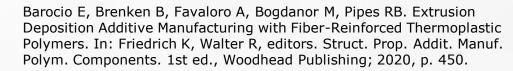
$$\frac{dX_{vc}}{dT} = k_{mb} \{ \exp^{[-k_{mb}(T-T_c)]} \} \cdot (1 + (d-1)\exp^{[-k_{mb}(T-T_c)]})^{\frac{d}{1-d}}$$

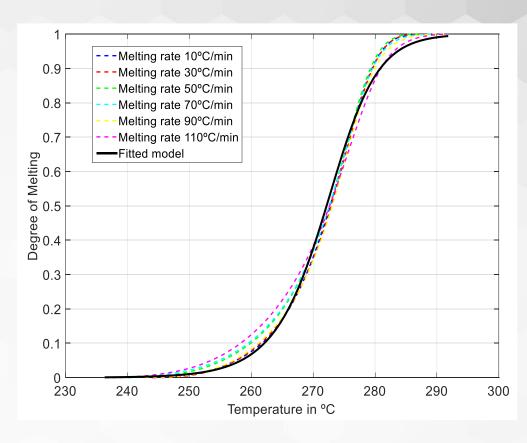
 T_c - crystallization temperature

- k_{mb} sharpness of the distribution of lamellar thicknesses d shape factor of the distribution of lamellar thicknesses
- γ_e lateral surface energy
- T_m° thermodynamic melting temperature

 ΔH_f - enthalpy of fusion

l - lamellar thickness







Thermoviscoelastic Material Behavior

Anisotropic linear constitutive equation:

$$\sigma_i = \int_0^t C_{ij}(T, X, t - \tau) \frac{\partial \varepsilon_j^{eff}}{\partial \tau} d\tau, \qquad i, j = 1, 2, \dots 6$$

T: Temperature, X: Crystallinity, t: Time

 $\varepsilon_j^{eff} = \varepsilon_j - \varepsilon_j^{inel}$: Effective strain

Time-Temperature-Superposition (TTS):

 $\xi_{ij}(t) = \int_0^t \frac{1}{a_T(T(\tau))} d\tau$

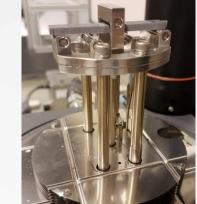
 $\xi_{ij}(t)$: Reduced time, $a_T(T)$: Shift factor (WLF Equation)

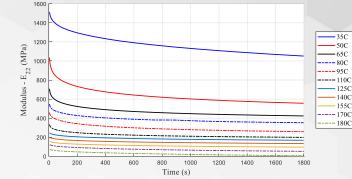
$$C_{ij}\left(T_0, X, \xi_{ij}(t) - \xi_{ij}'(\tau)\right) = f(X) \cdot \left[C_{ij0} + \sum_{w=1}^N C_{ijw} \exp\left(-\frac{\xi_{ij}(t) - \xi_{ij}'(\tau)}{\lambda_{ijw}}\right)\right]$$

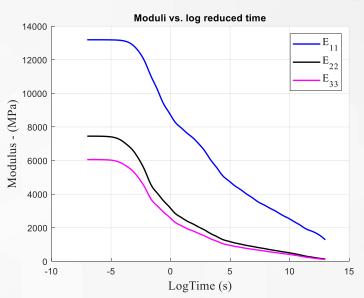
 λ_{ijw} : Relaxation times

f(X): Crystallization factor

 C_{ij0} , C_{ijw} : Relaxed/Unrelaxed stiffness parts



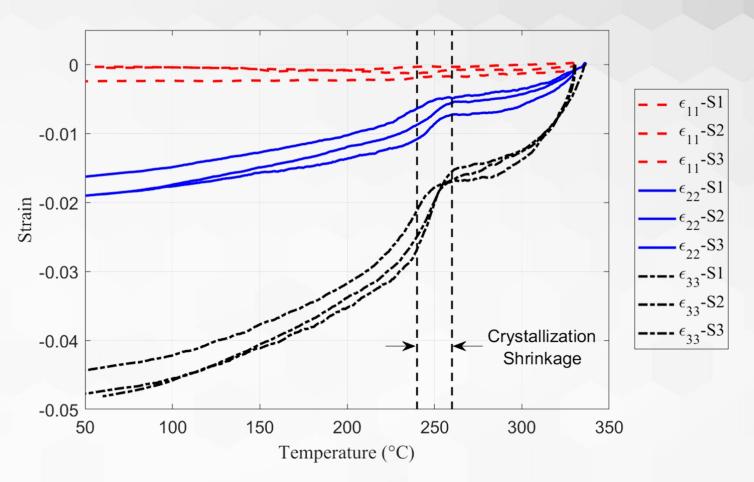




Brenken B, Favaloro A, Barocio E, Kunc V, Pipes RB. Thermoviscoelasticity in extrusion deposition additive manufacturing process simulations. 32nd Tech. Conf. Am. Soc. Compos. 2017, vol. 1, 2017.

Anisotropic Shrinkage

- Contribution from both thermoelastic and crystallization shrinkage
- Thermoelastic shrinkage: Governed by $\alpha(T)$
- Strain developed before crystallization does not contribute significantly to build up stresses



Barocio E, Brenken B, Favaloro A, Bogdanor M, Pipes RB. Extrusion Deposition Additive Manufacturing with Fiber-Reinforced Thermoplastic Polymers. In: Friedrich K, Walter R, editors. Struct. Prop. Addit. Manuf. Polym. Components. 1st ed., Woodhead Publishing; 2020, p. 450.

Interlayer and Fusion Bonding

Build Plate

- Non-isothermal fusion bonding model provides prediction of fracture properties
- Integrates effect of polymer crystallization on precluding reptation of polymer chains
- Model extended to inter-bead fusion bonding

$$D_b(t) = \frac{G_{1c}(t)}{G_{1co}} = \left[\int_0^{t_c} \frac{1}{t_w(T(\tau))} d\tau \right]^{\frac{1}{2}}, t_c \in \{t \ s. \ t. \ X_{vc} < X_{vc}^{crit}\}$$

$$t_w(T(\tau)) = A \cdot exp\left(\frac{E_A}{RT(\tau)}\right)$$

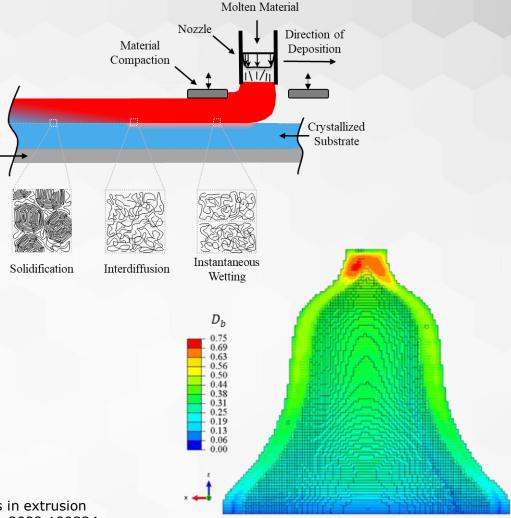
t_w - welding time

E_A - activation energy

A - pre-exponential factor

 $G_{1C\infty}$ - critical energy release rate of a perfectly bonded interface

Barocio E, Brenken B, Favaloro A, Pipes RB. Interlayer fusion bonding of semi-crystalline polymer composites in extrusion deposition additive manufacturing. Compos Sci Technol 2022:109334. https://doi.org/10.1016/j.compscitech.2022.109334.



7/16/2023



Interaction with Print Substrate

Adhesive / Beadboard



Boundary Conditions $\frac{\text{Thermal}}{q_s^{"}} = 0 | T_s = T_{bp} | R_{\theta}$

 $\frac{\text{Mechanical}}{\text{Cohesive Contact}}$ $t_n(\delta_1, D), t_s(\delta_s, D), t_t(\delta_t, D)$

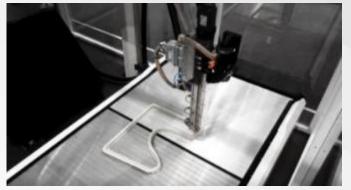
Build Sheets



Boundary Conditions $\frac{\text{Thermal}}{T_{bs} = T_{bp} \mid R_{\theta}}$

 $\frac{\text{Mechanical}}{\text{Spring Connectors}}$ $F_i = F_i^0 + \frac{D_{ij}u_j}{D_{ij}u_j}$

Rigid Constraint



Boundary Conditions $\frac{\text{Thermal}}{T_s = T_{bp} | R_{\theta}}$

 $\frac{\text{Mechanical}}{\text{All DOF Constrained}}$ $u_i = 0$

Barocio E, Thomas AJ, Pipes RB. Virtual Investigation of Residual Part Deformation Due to Build Plate Support Characteristics in Material Extrusion Additive Manufacturing. CAMX 2020 – Compos. Adv. Mater. Expo, VIRTUAL EXPERIENCE: CAMX 2020; 2020.



Material Card Characterization

Bead-Level Effective Properties

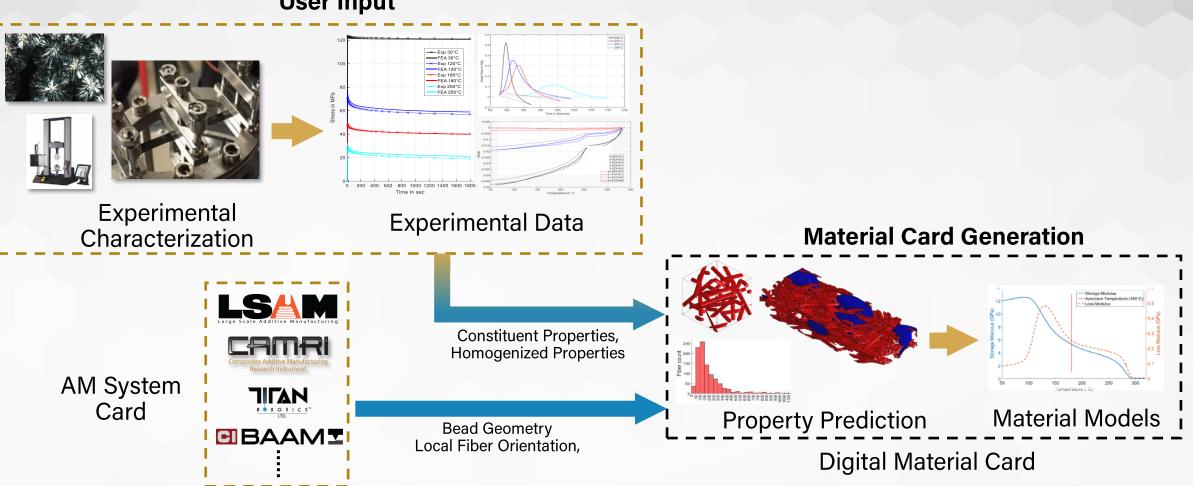


| Material Property | Relevant Standard |
|---|--------------------------|
| Glass Transition Temperature | ASTM D7028 |
| Fiber Orientation Distribution | |
| Fiber Length Distribution | |
| Elastic Properties (9 components) | ASTM D3039
ASTM D5379 |
| Coefficient of Thermal Expansion (3 directions) | ASTM E831 |
| Crystallization Kinetics and Melting | |
| Thermoviscoelastic Behavior (9 Prony series) | ASTM D5023 |
| Thermal Conductivity (3 directions) | ASTM E1461 |
| Heat Capacity | ASTM E1269 |
| Fusion Bonding Time | |

CM-C



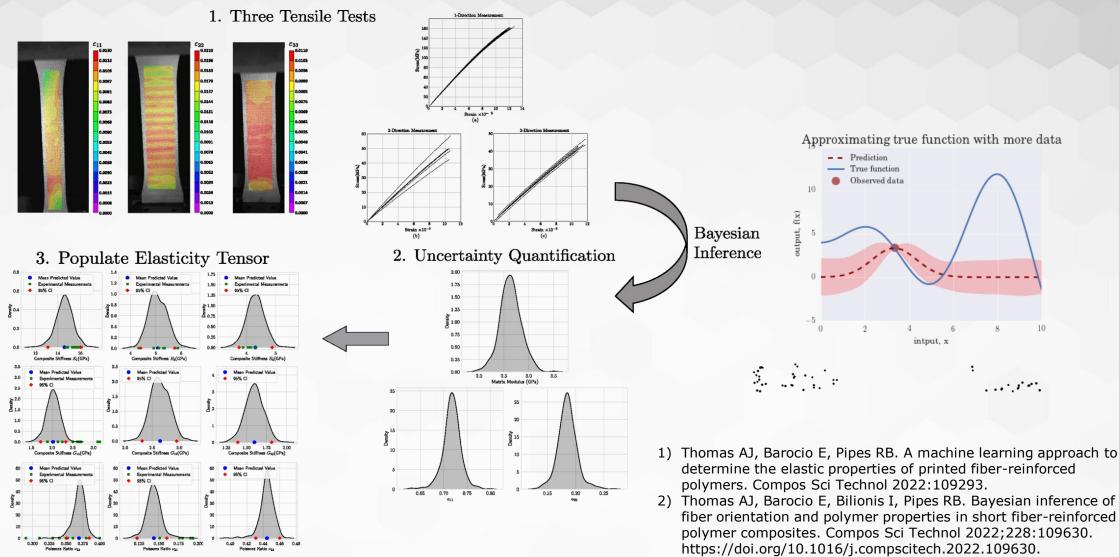
Material Card Generation



User Input

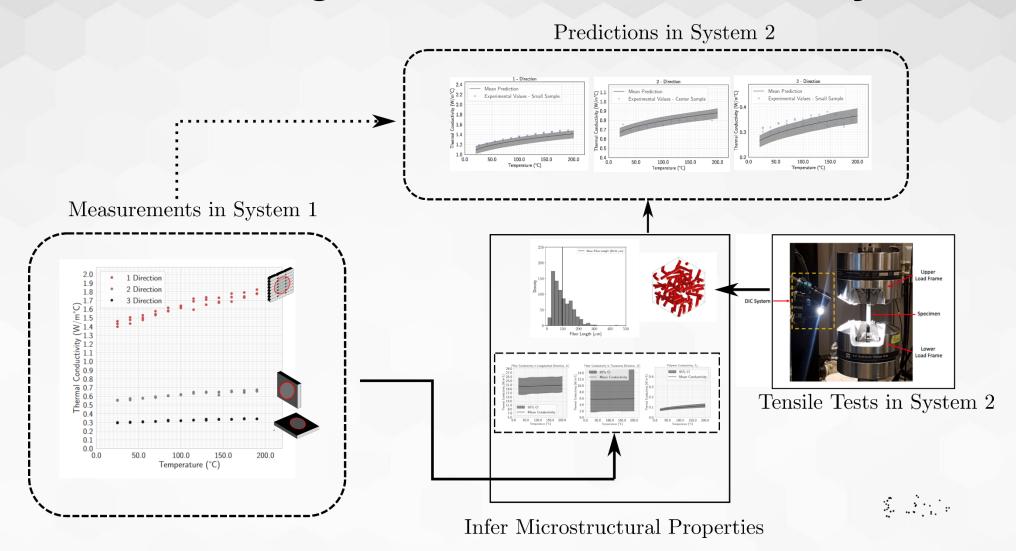


Accelerating Material Card Generation





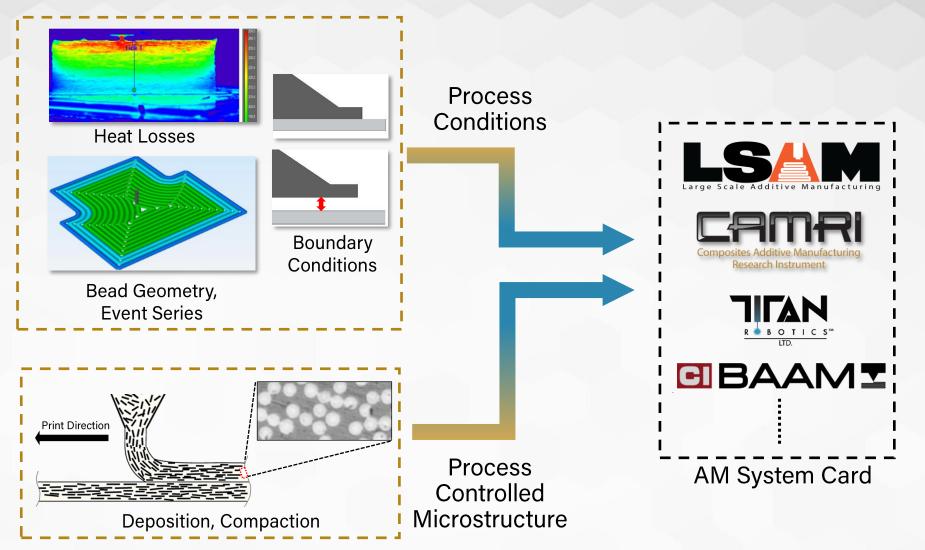
Transferring Material Cards Across AM Systems



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Machine Card Characterization

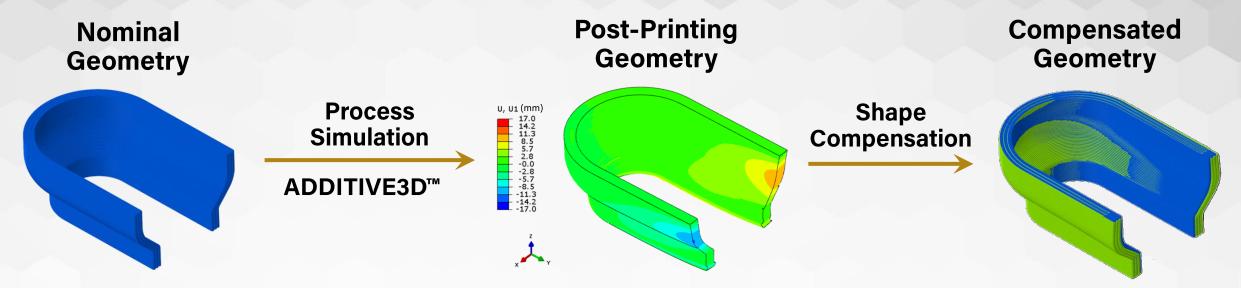




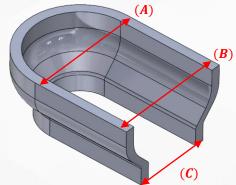
Simulation Driven Design with ADDITIVE3D



Simulation-Driven Shape Compensation of Autoclave Tool



Printed vs Nominal Geometry



| B) | Location | Nominal vs
Printed
Geometry
(mm) | |
|------------|----------|---|---|
| | Α | 0.71 | • |
| | В | 1.01 | |
| | С | 0.89 | |
| | | | |

Removal of Printed Geometry

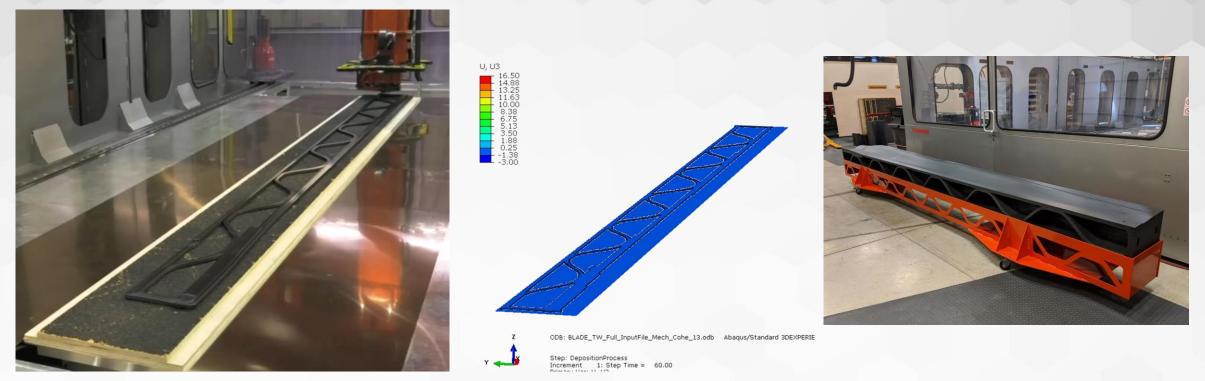




Validation



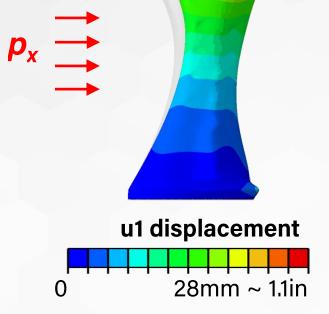
Reduction of Part Deformation Driven by Analysis



- Autoclave tool made from Techmer PM Electrafil[®] PESU 1810 3DP
- Three failed print iterations utilizing ~1,500lb each -> ~75% savings.
- ADDITIVE3D simulations utilized after multiple failures to drive success on the final print.



Structural Analysis of Printed Torch



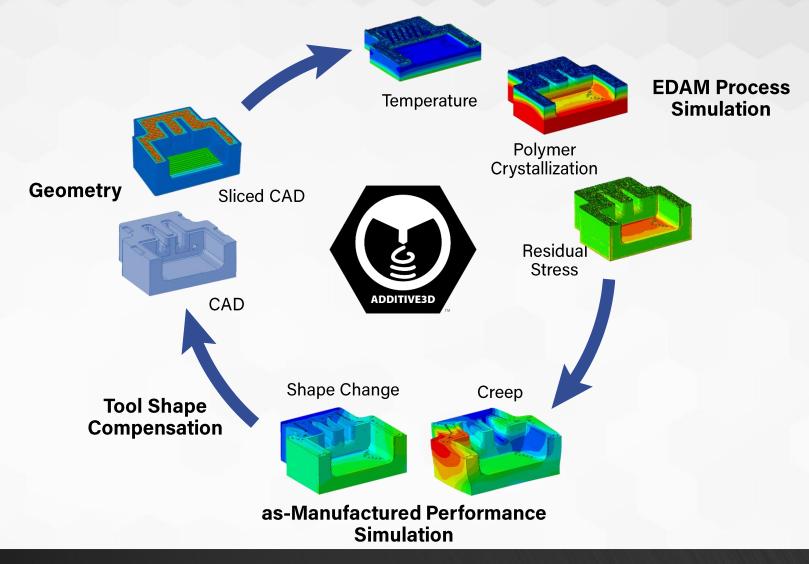




- Structural analysis to quantify displacement under load
- Risk assessment to ensure structural integrity



ADDITIVE3D for "First time right printing"







Horizons and Opportunities in Composites Additive Manufacturing



Printing of Hybrid Continuous and Discontinuous Fiber Systems

Opportunity

- Elevate the large-scale additive manufacturing process to produce semi-structural components
- Printing rates similar to large scale additive manufacturing with short fiber reinforced polymers are possible
- Selective printing of continuous and discontinuous fibers
- Integration of sensors and services (heating) in continuous fibers
- Process demonstrated in the CAMRI system at Purdue with continuous fiber filament impregnated in house (AS4 – PPS)

Tensile coupon printed with continuous and discontinuous fibers

Continuous fiber



(12) United States Patent Barocio et al. (10) Patent No.: US 11,214,006 B2 (45) Date of Patent: Jan. 4, 2022

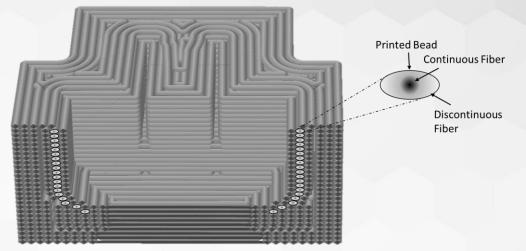
(54) METHODS AND APPARATUS FOR ADDITIVE MANUFACTURING UTILIZING MULTIFUNCTIONAL COMPOSITE MATERIALS, AND ARTICLES MADE THEREFROM

(58) Field of Classification Search CPC ... B29C 64/336; B29C 64/118; B29C 64/209; B33Y 10/00; B33Y 30/00; B33Y 40/00; B33Y 70/00 See application file for complete search history.



Printing of Continuous and Discontinuous Fibers

- Fundamental problems to investigate:
 - Concurrent flow of continuous and discontinuous fibers from nozzle to deposition
 - Adhesion between continuous and discontinuous fiber systems
 - Optimal microstructures for load transfer
 - Strength characteristics of hybrid printed materials
 - Design for printing with hybrid continuous and discontinuous fibers



Compression molding tool concept printed with continuous and discontinuous fibers



Compression Molding of SMC and Printed Continuous Fiber Preforms

Opportunity

- Increase strength characteristics of compression molded components
- Reduce variability in strength characteristics which elevates design allowables for this class of material systems
- Elevate value of recycled material (upcycle) with the addition of a small fraction of continuous fiber printed preforms
- Initial demonstration of the potential for this technology through a pin bracket geometry in collaboration with **S** 9T LABS



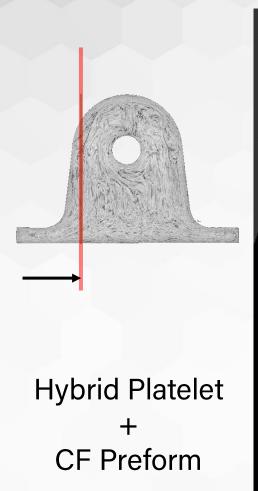
Configuration

Initial Continuous

Fiber Preform



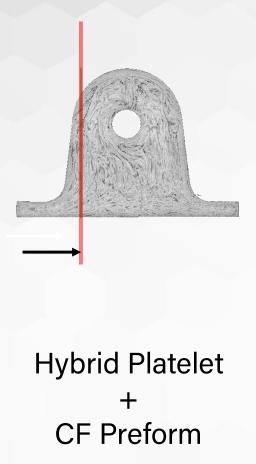
CT-Scan of Bracket Molded with Hybrid Material System

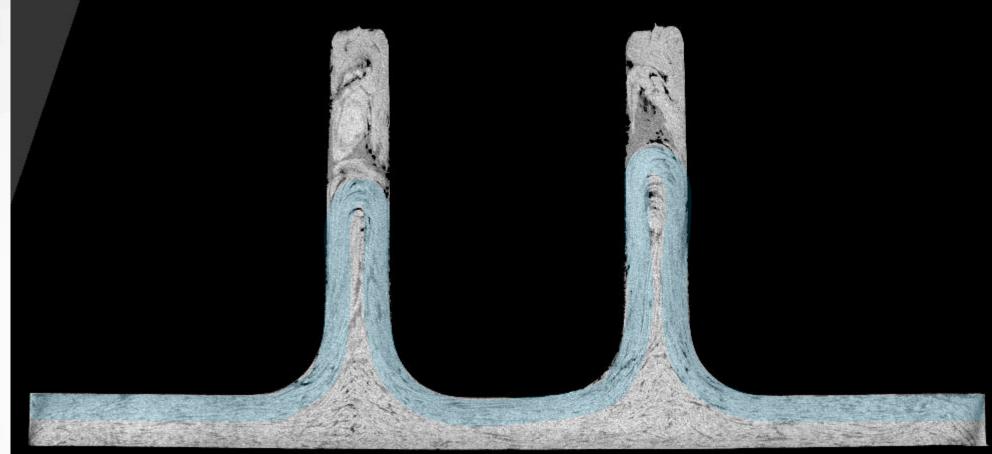






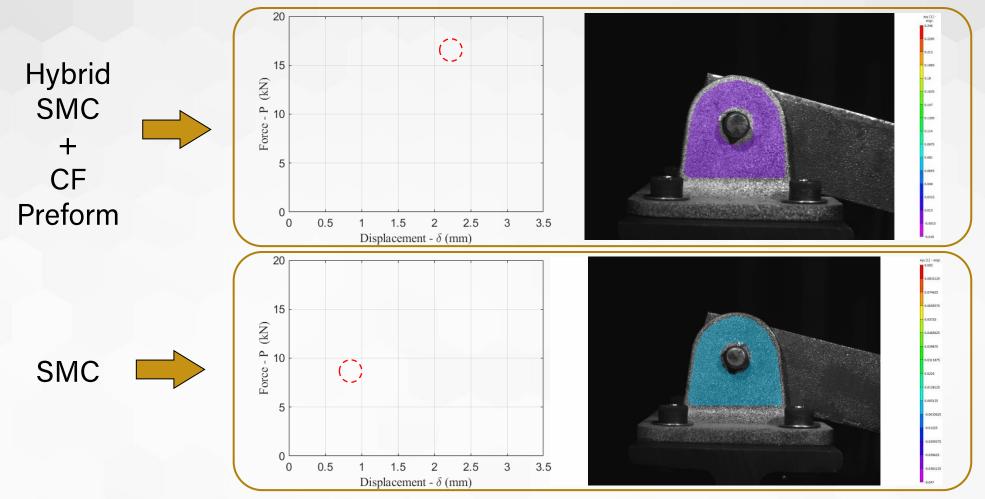
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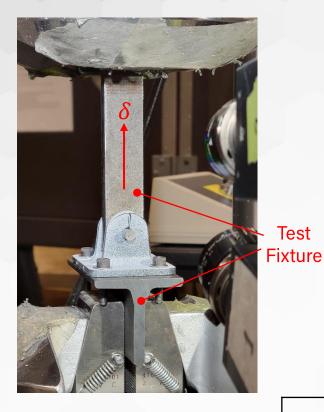
Structural Performance of Hybrid Material System

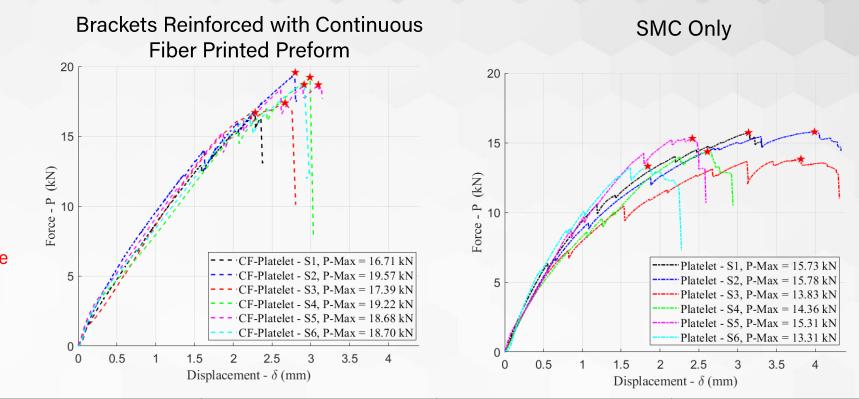


Barocio E, Eichenhofer M, Kalman J, Fjeld LM, Kirchhoff J, Kim G, et al. Compression Molding of Hybrid Continuous and Discontinuous Fiber Reinforced Thermoplastics for Enhancing Strength Characteristics. SAMPE Conf., Seattle, WA: SAMPE; 2023.



Compression Molding of Continuous and Discontinuous Fibers





| Pin Bracket Type | Average | Standard Deviation | Coefficient of Variance | | | |
|----------------------------------|-------------------|--------------------|-------------------------|--|--|--|
| Load - P (kN) @ Onset of Failure | | | | | | |
| SMC | <mark>7.67</mark> | 1.40 | 18.19% | | | |
| CF & SMC | 15.32 | 1.50 | <mark>9.81%</mark> | | | |
| Load - P (kN) @ Ultimate Failure | | | | | | |
| SMC | 14.72 | 1.04 | 7.05% | | | |
| CF & SMC | 18.38 | 1.10 | 6.01% | | | |



Acknowledgements

Composites Additive Manufacturing Group



Research Sponsors / Partners





Thank You